

AMENDMENTS TO THE SPECIFICATION:

Please replace the specification (translation) with the enclosed substitute specification. As required, both clean and marked-up versions are being submitted herewith. No new matter is believed to have been added.

Please replace the Abstract on page 21 of the specification (translation) with the following new Abstract:

A multilayer ceramic component includes a stack containing ceramic layers and electrode layers interspersed among the ceramic layers. The electrode layers contain copper and define first and second internal electrodes. First and second external contacts are on different sides of the stack. The first and second external contacts contain copper and are substantially perpendicular to the ceramic layers and electrode layers. The first internal electrode is connected to the first external contact and the second internal electrode is connected to the second external contact. The first and second internal electrodes overlap each other at a plane intersecting the stack. In areas adjacent to boundaries between the first and second external contacts and the ceramic layers, the first and second external contacts are not oxidized and material making-up the ceramic layers is not diminished. A bonding strength of the external contacts to the stack exceeds 50N.

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CERAMIC MULTI-LAYER COMPONENT AND
METHOD FOR THE PRODUCTION THEREOF

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Technical Field

This application relates to a multilayer ceramic component and a method for producing the component. The component includes a stack of alternating ceramic layers and copper-containing electrode layers, which serve as internal electrodes. The internal electrodes are connected to external contacts that are arranged on opposite, exterior sides of the stack, perpendicular to the multilayer structure. Internal electrodes that are connected to different external contacts are in interlocking engagement with one another.

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Background

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A component of foregoing type and a method for its production are known, for example, from DE 20023051.4.

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DE 9700463 describes a method for producing green films for multilayer piezoceramic components with Ag/Pd internal, in which a PZT-type piezoceramic powder (PZT = lead zirconate titanate) is used.

The material and/or the process used to fire on the external electrodes that contact the internal electrodes should, in principle, be selected such that the electrode metal will

not oxidize and the ceramic will not be reduced. For this reason, a precious metal or a precious metal alloy is customarily used as the electrode material.

DE 19945933 describes an exemplary method for producing external electrodes in piezoceramic components comprised of PZT ceramic and Ag/Pd internal electrodes. In this method, contacting of the Ag/Pd internal electrodes is accomplished using a metal paste that has a silver content of > 65 % and an organic binder, which is fired on at approximately 700° C. The firing on of the metal paste is conducted in an air atmosphere, because aromatic compounds contained in the organic paste binder cannot be completely degraded under reductive conditions. However, this method is not suited to a multilayer component with PZT ceramic and copper-containing internal electrodes, since at customary debinding and/or firing-on temperatures, reduction of the PZT ceramic and oxidation of the metallic copper are prevented under only one specific, very low oxygen partial pressure of $< 10^{-2}$ Pa. Thus, the Ag/Pd external electrodes cannot be used in a multilayer ceramic component with copper-containing internal electrodes.

DE 20023051 U1 describes a component that is less costly than multilayer piezoceramic components that are based on a PZT ceramic and Ag/Pd internal electrodes. This component uses copper-containing internal electrodes in place of the expensive Ag/Pd internal electrodes. However, DE 20023051 U1 does not indicate which material would be suitable for firing on external electrodes in a piezoceramic component with copper-containing internal electrodes.

Summary

Described herein is a multilayer ceramic component with internal electrodes and external contacts for contacting the internal electrodes, in which the bonding strength of the external contacts is sufficient. Also described is a method for producing the component, in which the ceramic is not reduced and in which neither internal nor external electrodes will oxidize.

The ceramic component of this application contains a stack of alternating ceramic layers and copper-containing electrode layers that serve as internal electrodes. The internal electrodes are connected to external contacts. The external contacts are arranged on exterior surfaces of the stack, opposite one another and perpendicular to the multilayer structure. The internal electrodes that are connected to different external contacts are in interlocking engagement with one another. The external contacts contain metallic copper. The bonding strength of the external contacts to the stack is at least 50 N.

The ceramic layers may be comprised of ceramic green films that contain a thermohydrolytically degradable binding agent and can comprise a ferroelectric perovskite ceramic of the general composition ABO_3 , such as one of the PZT type $Pb(Zr_xTi_{1-x})O_3$.

The application further describes a method for producing a ceramic component, in which binder removal, or debinding, is performed and/or completed at a temperature of

5 $\leq 300^{\circ}\text{C}$. The debinding takes place under a nitrogen stream with the addition of water vapor. The water vapor partial pressure is set such that the corresponding oxygen partial pressure at the given temperature is between the equilibrium points for $\text{Cu}/\text{Cu}_2\text{O}$ and PbTiO_3/Pb . The equilibrium point corresponds to an oxygen partial pressure at which both a reduced metal and a metal compound that corresponds to this metal are thermodynamically stable and can coexist without diffusing into one another.

10 Paste binders are completely separated out in a reduced atmosphere of, for example, $< 10^{-2}\text{ Pa}$ at a comparatively low temperature of $\leq 300^{\circ}\text{C}$, because at higher debinding temperatures, the oxygen, which is insufficient to burn off the carbon contained in the organic binder, is partially drawn out of the ceramic lattice structure, which impairs properties of the ceramic layers.

15 Complete separation of organic constituents occurs because debinding is performed in a nitrogen stream that is charged with water vapor, resulting in hydrolytic separation. The addition of water vapor causes the oxygen partial pressure to decrease thermodynamically. However, the oxygen partial pressure does not drop below a specific level at which the ceramic would begin to reductively degrade.

20 On the other hand, the oxygen partial pressure also will not exceed a level at which metallic copper would begin to oxidize at the given temperature. Thus the oxygen partial pressure is selected to be low enough so that the reduction processes in the ceramic is

adequately inhibited, while at the same time the copper contained in the metal paste will not oxidize.

The oxygen partial pressure is maintained, or correspondingly adjusted, at all times to correspond to the temperature. This occurs not only during the debinding process, but also during firing on of the metal paste, such that in a p(T) diagram it lies between the equilibrium points for Cu/Cu₂O and Pb/PbO at each process temperature.

The proportion of copper in the metal paste may be > 70%. Acrylic resin binders may be used as the organic paste binders.

The foregoing makes it possible to use a (copper-containing) metal paste with an organic paste binder to produce external contacts in a piezoceramic component with internal electrodes that contain copper.

To produce the external contacts, a copper-containing metal paste having a copper content of > 70 m%, for example 78 m%, a glass flow, and an organic binder, for example acrylic resin binder, may be used.

Glass flow (glass frit) may be comprised essentially of PbO and SiO₂, but may also contain other components, such as Na₂O, Al₂O₃ and BaO. The proportion of glass flow in the metal paste may be less than 5 m%. The composition and the proportion of the

glass flow are selected such that the glass frit contained in the metal paste of the external contacts partially diffuses into the ceramic, thereby increasing the bonding strength of the external contacts to the sides of the stack.

5 First, using a known process produces a stack comprised of layers of ceramic material and electrode layers placed one on top of another, which in a finished component correspond to the internal electrodes. The electrode layers are comprised of a copper-containing metal paste and can be applied to the layers made of a ceramic material, for example, via screen printing.

10 The copper-containing metal paste is also applied, for example, via screen printing, on opposite sides of the stack of internal electrodes and ceramic layers, which are positioned one on top of another.

15 The binder is removed from the metal paste in a moist nitrogen atmosphere in a gas-tight furnace at a temperature of $\leq 300^{\circ}\text{C}$, after which it is sintered at a higher temperature.

The copper-containing metal paste may be fired on at between 700 and 860°C .

To prevent the reduction of the lead oxide contained in the ceramic, lattice bases made of metallic copper, which serve simultaneously as gettering material, can be used in production of a component described herein.

5 Below, the component, and method for its production, are described in greater detail with reference to exemplary embodiments and the associated drawings. The figures contain schematic, not-to-scale representations of various exemplary embodiments. Identical or equivalent components are assigned the same reference figures.

10 Description of the Drawings

Figure 1 is a perspective view of the structural design of a ceramic multilayer component.

Figure 2 is a half-logarithmic representation of oxygen partial pressure, which can be adjusted via the addition of water vapor, as a function of the temperature and the
15 equilibrium curves for Cu/Cu₂O and Pb/PbTiO₃.

Detailed Description

Figure 1 shows a ceramic component with copper-containing external contacts AK1 and AK2. First copper-containing internal electrodes IE1 are connected to the first
20 external contact AK1. Second copper-containing internal electrodes IE2 are connected to the second external contact AK2. The internal electrodes are separated from one another by ceramic layers KS.

The ceramic layers KS may possess piezoelectric properties and may be produced using the PZT-type ceramic.

5 The component of Figure 1 is a piezoelectric actuator. The electrode layers and ceramic layers stacked one on top of another are referred to herein as a piezo stack.

 The fired-on external contacts AK1, AK2 may be between 10 and 20 μm thick, e.g., 15 μm , thick. However, a different thickness for the external contacts may also be
10 used.

 The external contacts and/or internal electrodes may contain a certain proportion of ceramic, which may be less than 50 m% (m% - percentage by mass), and which in one embodiment is between 10 and 50 m%, e.g., 40 m%. In this embodiment, the ceramic
15 proportion comprises ceramic particles having a specific grain size, for example, an average grain size of between 0.2 and 0.6 μm .

 The ceramic portion in the metal paste serves to inhibit the formation of cracks and separation of the external contact from the piezo stack, which can result from different
20 expansion properties of the ceramic material and metallic copper.

The thermal expansion coefficient of PZT-type ceramic within the temperature range extending from room temperature to the Curie temperature is approximately 1.5 - 2.0 ppm/K. In the same temperature range, metallic copper has a substantially higher thermal expansion coefficient of approximately 19 ppm/K. By admixing the ceramic material into the metal paste, the expansion characteristics of the external contact are adjusted to the expansion behavior of the ceramic stack both for the processing of the component and for its later applications within the specified temperature range of, for example, -50° C to +150° C, wherein, for example, the application of an electrical field causes a deformation of the component.

To adjust the thermal expansion coefficient, a glass frit that contains a high proportion of SiO₂, for example 39 m%, may be used, since SiO₂ has a high affinity for PZT-type ceramic.

In preparation of the copper-containing metal paste, first a ceramic powder having an average grain size of, for example, 0.4 µm is dispersed in a solvent. The ceramic powder dispersion is then stirred into the copper-containing metal paste of the above-named composition, and is homogenized using a three-roller drawing machine. The viscosity of the metal paste may be between 10 and 20 Pas. Once the finished metal paste has been applied to the side surfaces of the piezo stack, the paste is dried at approximately 80 to 140° C in an air atmosphere. This is followed by debinding and sintering under the foregoing specified conditions, in which oxidation of metallic copper and reduction of

PbO or PbTiO₃ are prevented. In this embodiment, selection of the debinding temperature and the duration of the debinding process ensures that, during debinding, both the binder components and the residual solvent will be completely burned out of the metal paste.

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Because the glass frit contained in the metal paste can diffuse quite extensively into the ceramic, leaving behind hollow cavities in the sintered ceramic layers, the sinter temperature is selected to be low enough (for example, 765° C) that the glass additive can penetrate only in the area of the internal electrodes. Microscopic studies have shown that at such a sinter temperature glass portions, above all silicon oxide, can be detected within a narrow area of the ceramic layers that is adjacent to the external contacts. The external contacts adhere firmly and can be detached from the piezo stack with the application of a high level of force, greater than 50 N. With the forceful detachment of the external contacts, portions of the ceramic material are broken off, which indicates a high level of bonding strength for the external contacts on the piezo stack.

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The ceramic proportion may be 40 m% relative to the solids content of the metal paste. This metal paste can also be used for internal electrode layers.

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A chemically active ceramic powder (or even some other chemically active additive), hereinafter also referred to as a ceramic additive, may be present in the electrode metallization, which under certain circumstances can react chemically with the

electrode metal, the organic binder, the ceramic, and/or a reaction product of the latter, or can chemically bind certain components. Furthermore, the ceramic additive can react with the process atmosphere, for example, it can release oxygen into the process atmosphere or can absorb oxygen so that the oxygen partial pressure is stabilized at least locally or temporarily. With a stable oxygen partial pressure, it is possible to protect internal electrode layers or external contacts from oxidation and to protect ceramic layers from reduction. The ceramic powder additive ensures that a metal oxide that has been formed as a result of process instabilities in the electrode metallization is bonded, thus preventing an undesirable diffusion of this metal oxide into the ceramic layers.

The process of using a chemically inert ceramic powder in a metal paste, for example, to retard the sintering of the metal, is known in the art. In this application, however, the ceramic powder is used as a functional additive that is chemically active and can chemically react with its environment. The chemical activity can be focused, for example, on binding of Pb, which is released from the lead-containing ceramic material during sintering. It is also possible for the chemically active ceramic powder to bind another component from the ceramic mass especially during sintering, or to promote the release of certain components, such as oxygen from the ceramic mass or from the binder contained in the ceramic mass or in the metal paste. The ceramic powder should not chemically react with the metal portion of the metal paste.

In contrast to the metal paste that is used for the external electrodes, in the metal paste that is suitable for use in the internal electrode layers, preferably no glass additives are used. $(\text{Zr,Ti})\text{O}_2$ is particularly well suited for use as the chemically active ceramic powder. In place of a chemically active ceramic powder, the metal paste can also contain
5 another chemically active additive, or in addition to the chemically active ceramic powder it may contain portions of other substances, for example BaO_2 and/or MgO .

The addition of a ceramic powder in the case of the internal electrodes further serves to improve adhesion between the internal electrodes and the ceramic layers that
10 surround them. A fine distribution of the ceramic particles among the metal particles especially serves to prevent sinter neck formation. The sinter necks represent a localized disconnection of the internal electrodes, in which the metal coating becomes detached from the ceramic layer and/or becomes pulled back - especially around the edges - so that the internal electrodes assume a net-like structure, which is not replicable from component
15 to component. The method of achieving a homogeneous internal electrode structure by adding a precious metal or a precious metal alloy is known in the art. However, the addition of the ceramic powder as described herein also provides a substantial cost advantage over the known method.

20 Figure 2 shows a half-logarithmic representation of an oxygen partial pressure p_{O_2} that can be adjusted based upon the temperature - by adding water vapor - and is

illustrated by way of example by the curve 3, the numerically calculated equilibrium curve 1 for Cu and Cu₂O, and the numerically calculated equilibrium curve 2 for Pb and PbTiO₃.

The equilibrium curve 1 indicates the partial pressure of O₂ at the selected temperature, at which metallic Cu and Cu₂O can coexist. Metallic copper exists only at an oxygen partial pressure p_{O_2} that does not exceed the equilibrium level, i.e. below the equilibrium curve 1. Because only Cu₂O is stable above the curve 1, at an oxygen partial pressure that exceeds the equilibrium value at the selected temperature an undesirable oxidation of the metallic copper will occur.

The equilibrium curve 2 indicates the partial pressure for O₂ at the selected temperature, at which metallic Pb and PbTiO₃ can coexist. PbTiO₃ exists only above the curve 2. At an oxygen partial pressure p_{O_2} that drops below the equilibrium level at the selected temperature, the PbTiO₃ contained in the ceramic will be reduced to Pb.

For this reason, at least during the debinding phase of a process for producing the above-described multilayer ceramic component, the oxygen partial pressure p_{O_2} is adjusted by adding water vapor such that the oxygen partial pressure does not exceed the maximum level p_{\max} as defined by the curve 1, at which metallic copper is still stable, but also will not drop below the minimum level p_{\min} as indicated by the curve 2, at which lead titanate is not yet reduced, i.e., $p_{\min} < p_{O_2} < p_{\max}$ at the given debinding temperature T_E .

The permissible range for adjustment of the oxygen partial pressure is between the curves 1 and 2.

5 Curve 3 shows an optimal oxygen partial pressure to be established based upon temperature and within a moist atmosphere. The quantity of water vapor to be added can be calculated in principle from the curve 3. It is also possible to manually or automatically control the drop in the oxygen partial pressure by adding water vapor, in such a way that the preset threshold values are not exceeded.